## **PHOTONICS Research**

# Superior third-order nonlinearity in inorganic fullerene-like WS<sub>2</sub> nanoparticles

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Two-dimensional (2D) transition metal dichalcogenides (TMDs) attain increasing attention due to their exceptional nonlinear optical efficiencies, which hold promising potential for on-chip photonics and advanced opto-electronic applications. Planar TMDs have been proven to support orders-higher third-order nonlinear coefficients in comparison with common nonlinear materials. Interestingly, stronger light-matter interaction could be motivated when curved features are introduced to 2D TMDs. Here, a type of inorganic fullerene-like WS<sub>2</sub> nanoparticles is chemically synthesized using hard mesoporous silica. By using the spatial self-phase modulation (SSPM) method, the nonlinear refractive index  $n_2$  and third-order susceptibility  $\chi^{(3)}$  are investigated in the visible range. It is found that  $n_2 \sim 10^{-5}$  cm<sup>2</sup>/W and  $\chi^{(3)} \sim 10^{-7}$  esu, two orders higher than the counterparts of planar WS<sub>2</sub> structures. Our experimental findings provide a fresh thinking in designing nonlinear optical materials and endow TMDs with new potentials in photonic integration applications. © 2020 Chinese Laser Press

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### 1. INTRODUCTION

Nonlinear optical effects ignite exciting light-matter interactions and greatly enlarge optical applications such as frequency conversion, optical imaging, and information processing [1,2]. Remarkable achievements have been made based on novel working principles [3] and materials [4,5] and their marriage [6–8]. However, one of the main challenges hindering the full exploration of nonlinear effects is the low intrinsic nonlinear susceptibilities of conventional materials. Over the past decade, two-dimensional (2D) materials have attracted increasing attention due to their outstanding optical, electronic, and mechanical properties [9–12]. As a typical example, transition metal dichalcogenides (TMDs) possess layer-dependent electronic bandstructure and thus tunable linear and nonlinear optical properties [13,14]. In particular, third-order nonlinearity holds unique importance for applications in mode-locked lasers, sensors [15], and all-optical switching and modulation [16]. To characterize the third-order susceptibility, various methods have been proposed, such as Z-scan [17,18], four-wave mixing [19], and spatial self-phase modulation (SSPM) [20].

So far, the exploration of optical properties of TMD materials has been mainly focused on various flat 2D structures, including nanosheets or nanoflakes [21–23]. However, in

addition to the size, shape, thickness, and material quality of TMDs, the geometric characteristics are also supposed to greatly affect their optical properties [24]. In contrast, inorganic fullerene-like (IF-like) 2D nanoparticles (NPs) with curved geometric features introduce an additional freedom to control and enhance the light–matter interaction strength [25,26]. Initially, they are widely investigated as an efficient lubrication material [27,28]. Recently, it was found that the curved features are prone to symmetry breaking to 2D materials and then making the silent phonon mode Raman active [26,29,30]. It is thus naturally speculated that other nonlinear effects may be enhanced with the curved features.

Here, IF-like WS<sub>2</sub> NPs are chemically synthesized using hard mesoporous silica. The nonlinear refractive index  $n_2$  and third-order susceptibility  $\chi^{(3)}$  are characterized using the SSPM method in the visible range. It is found that the nonlinear optical responses of the proposed structures are orders stronger than the counterparts of planar 2D WS<sub>2</sub> films. Therefore, we believe that curved 2D materials could play a growing role in designing optical materials with superior efficiencies at each order of nonlinearity and are endowed with new potentials in high-speed optical signal processes and photonic integration applications.

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#### 2. EXPERIMENT

The IF-like WS<sub>2</sub> NPs are chemically synthesized using ordered three-dimensional (3D) mesoporous silica (EP-FDU-12) as hard templates. The average diameter of the pores and thickness of the wall are 27 nm and 5 nm, respectively. The precursor, i.e., phosphotungstic acid (PTA), is incorporated into the template via a solvent evaporation process. The WS<sub>2</sub> NPs can then be obtained by removing the template in H<sub>2</sub>S gas. A typical scanning electron microscopy (SEM, JEOL, JSM-7000F) image of the synthesized WS<sub>2</sub> NPs is shown in Fig. 1(a). The multilayer structure with an interlayer distance of ~0.67 nm is clearly characterized using a high-resolution transmission electron microscopy (HRTEM, JEOL, JEM-2100F) image [Fig. 1(b)]. The synthesized WS<sub>2</sub> NPs hold IF-like features with an average diameter of ~26.5 nm, which do not exhibit a quantum size effect [25]. It is obvious that the NPs show curved multilayered features with a layer number >5 [Fig. 1(b)]. The peaks in the X-ray diffraction (XRD, Bruker, D8 Advance) pattern match well with the standard WS<sub>2</sub> structure (JCPDS card No: 08-0237) [Fig. 1(c)].

Two Raman peaks are observed at  $\sim 353$  and  $\sim 420$  cm<sup>-1</sup> under the excitations of 532 and 633 nm continuous-wave (CW) lasers [Fig. 1(d)]. The Raman active lattice vibrations at the  $\Gamma$  point of the hexagonal Brillion zone are modes 421 cm<sup>-1</sup> and 356 cm<sup>-1</sup> in the detected region [31,32]. Furthermore, second-order Raman transition, i.e., two longitudinal acoustic (2LA) phonons at  $\sim 353$  cm<sup>-1</sup>, are also observed for excitation energies close to the band gap. More interesting is the activation of the B<sub>1u</sub> mode, which is silent in planar 2D TMDs. Its excitation arises from the curved layers and structural disorder of WS<sub>2</sub> NPs [26,29,30].

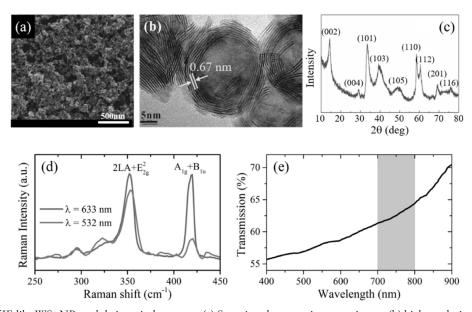
Under the illumination of an incoherent white light source, the transmittance was obtained by normalizing the transmitted power of the ethanol solutions with WS<sub>2</sub> NPs to that without

 $WS_2$  NPs. Figure 1(e) shows the transmittance spectrum of  $WS_2$  NP dispersion ranging from 400 to 900 nm (Andor SR500I), which is used to characterize the effective number of  $WS_2$  layers in the SSPM experiment. There is no evident excitonic resonance feature in the transmission spectrum, which may be attributed to the decrease in the exciton binding energy due to the increase in the number of  $WS_2$  layers [33].

The experimental setup for SSPM is schematically shown in Fig. 2(a). A femtosecond (fs) pulse laser (Coherent, Chameleon Ultra II, repetition frequency 80 MHz, pulse width 100 fs at 800 nm) propagates along the z axis and is loosely focused on the cuvette by a lens with a focal length of 200 mm. In the experiment, the incident power can be controlled using a set of neutral density (ND) filters. Then, diffraction patterns are recorded using a digital camera with a slow-motion function. Due to the SSPM effect, the transmitted light appeared as a set of conical shells, which form concentric rings on a 2D screen (Fig. 2). The outermost ring stripe is always brighter and wider than the inner ones. Interestingly, the initial concentric diffraction rings deform quickly [Fig. 2(b)]. The upper half of the ring pattern continuously collapses towards the center of the initial concentric rings and then enters a stable state. In contrast, the lower part distorts slightly. The evolution time from the generation of ring-shaped patterns to saturation of distortion phenomenon usually lasts from less than one second to several seconds, which relies on the impinging power.

#### 3. RESULTS AND DISCUSSION

Generally, the SSPM phenomenon exhibits as a series of concentric diffraction rings on a projection screen when a high-intensity laser beam interacts with the nonlinear medium. The SSPM ring pattern is attributed to the laser-induced refractive index change  $\Delta n$  [34]. As the laser beam propagates along the z axis, the field E reorients the direction of WS $_2$  NPs in the



**Fig. 1.** Structure of IF-like WS $_2$  NPs and their optical response. (a) Scanning electron microscopy image, (b) high-resolution transmission electron microscopy image, and (c) X-ray diffraction pattern of the synthesized IF-like WS $_2$  NPs. (d) Raman spectra of the WS $_2$  dispersion excited by 633 and 532 nm lasers. (e) Transmittance of WS $_2$  NP dispersions. The interesting wavelength range is highlighted in the gray area.

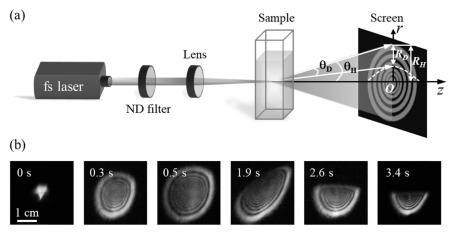


Fig. 2. (a) Schematic of the experimental setup and (b) evolution of the concentric ring-shaped diffraction patterns excited by a fs pulse laser at  $\lambda = 800$  nm. The time capturing the diffraction patterns is inserted at the upper-left corner of each image.

normal plane. According to the Kerr effect, the refractive index of the suspension can be described by  $n = n_0 + n_2 I$ , where  $n_0$  is the linear refractive index,  $n_2$  is the nonlinear refractive index of WS<sub>2</sub> NPs, and I stands for the incident intensity of laser beam [1]. It should be noted that the self-focusing effect occurs when the beam enters into the Kerr media. The beam size rapidly converges into a minimum after a propagation length of less than one millimeter. Then, the beam propagates like a plane wave with a slightly increased diameter due to weak absorption and light scattering. Therefore, the self-focusing effect usually is not taken into account when measuring the nonlinear refractive index using SSPM [20].

After traversing the  $WS_2$  dispersions of a thickness L, the incident light will gain an intensity-dependent phase [34]

$$\Delta\varphi(r) = \left(\frac{2\pi n_0}{\lambda}\right) \int_0^{L_{\text{eff}}} n_2 I(r, z) dz, \tag{1}$$

where I(r,z) is the intensity distribution of the focused laser beam,  $r \in [0, +\infty)$  is the transverse coordinate in the beam, and the host solvent is ethanol with a refractive index of  $n_0 = 1.36$ .  $L_{\text{eff}}$  represents the effective interaction length contributing to the SSPM process, which can be calculated by  $L_{\text{eff}} = \int_{L_1}^{L_2} (1 + z^2/z_0^2)^{-1} dz = z_0 a \tan(z/z_0) \Big|_{L_1}^{L_2}$ , where  $z_0 =$  $\pi\omega_0^2/\lambda$ , is defined by the waist width  $\omega_0$  and wavelength of the laser beam;  $L = L_2 - L_1$  is the thickness of the quartz cuvette. In the experiment, L=10 mm and  $\omega_0=74.2$  µm at the front surface of the cuvette. For simplicity, the incident Gaussian laser with a cylindrical symmetry along the z axis will gain an additional phase shift  $\Delta \varphi(r) = \Delta \varphi_0 \exp(-2r^2/\omega_0^2)$ after passing through the WS<sub>2</sub> dispersions. Here,  $\Delta \varphi_0$  is the phase shift at the diffraction ring center, i.e., r = 0 [34]. By using Eq. (1), we obtain  $\Delta \varphi_0 = 2\pi n_0 n_2 L_{\rm eff} I/\lambda$  with I(0,z) = 2I [35], which indicates that a larger intensity results in more phase shift. Since the temporal slot between pulses is 12.5 ns, all of the interference arises from the SSPM within each single pulse. Radiation fields from the area around two different points have the same wave vector and can cause interference. Maximum constructive or destructive interference is determined by  $\Delta \varphi(r_1) - \Delta \varphi(r_2) = m\pi$ , where m is an odd or

even integer corresponding to dark or bright stripes, respectively. The total number of diffraction rings can be estimated as  $N = [\Delta \varphi(0) - \Delta \varphi(\infty)]/2\pi = \Delta \varphi_0/2\pi$ , which linearly increases as laser intensity increases [Fig. 3(a)]. In addition, at a given incident intensity, more rings pour out at longer wavelength irradiation.

The nonlinear refractive index can be expressed as [20,23]

$$n_2 = \frac{\lambda}{2n_0 L_{\text{eff}}} \frac{\text{d}N}{\text{d}I}.$$
 (2)

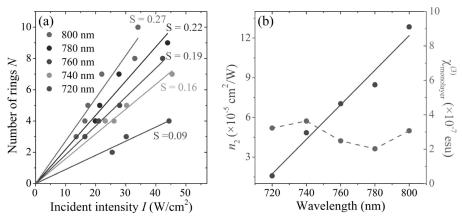
The slope S = dN/dI can be readily obtained by fitting intensity-dependent ring numbers, which increases as wavelength increases at a given intensity [Fig. 2(a)]. Moreover, the total third-order susceptibility can be obtained,  $\chi^{(3)}_{\text{total}} = \frac{\lambda c n_0}{2.4 \times 10^4 \text{ km}^2 L_{\text{eff}}} S$  [20,23,36].

As introduced previously, third-order nonlinear susceptibility can be obtained by the susceptibility can be readily obtained by fitting intensity of the susceptibility can be readily obtained by fitting intensity obtained by the susceptibility can be readily obtained by fitting intensity of the susceptibility can be obtained by fitting intensity of the susceptibility can be obtained by fitting intensity of the susceptibility can be obtained by fitting intensity of the susceptibility can be obtained by the suscepti

As introduced previously, third-order nonlinear susceptibility  $\chi^{(3)}$  is of great significance for indicating nonlinear performance of the nonlinear materials. Here, the third-order nonlinear susceptibility of monolayer WS<sub>2</sub> NPs can be estimated using the counterpart of multiple layer structures with  $\chi^{(3)}_{\text{total}} = N_{\text{eff}}^2 \chi^{(3)}_{\text{monolayer}}$  [20,23], where  $N_{\text{eff}}$  represents the effective number of WS<sub>2</sub> layers in the NPs, and  $\chi^{(3)}_{\text{monolayer}}$  represents the contribution of one layer WS<sub>2</sub> out of  $N_{\text{eff}}$  layers to the third-order susceptibility of WS<sub>2</sub> NPs. Therefore,  $\chi^{(3)}_{\text{monolayer}}$  can be calculated with the following equation:

$$\chi_{\text{monolayer}}^{(3)} = \frac{n_0^2 n_2(\text{cm}^2/\text{W})}{0.0395 \times N_{\text{eff}}^2}.$$
 (3)

The transmission of monolayer WS<sub>2</sub> is 99.3%–99.7% at the selected wavelength [23,37]. Therefore, according to the transmission measurement in Fig. 1(e), the effective layer number  $N_{\rm eff}$  is estimated to be 48–140 at the wavelength ranging from 720 to 800 nm. Thus, the third-order susceptibility  $\chi_{\rm monolayer}^{(3)}$  for monolayer WS<sub>2</sub> is estimated to be in order of ~10<sup>-7</sup> esu, which is two orders higher than the counterparts of popular 2D materials with planar features. Similar results are obtained when the solvent is replaced by methylbenzene. In addition, the  $n_2$  of ethanol is ~9 orders smaller than the counterpart of the WS<sub>2</sub>



**Fig. 3.** (a) Dependence of the number of SSPM rings N on the laser intensity I at different wavelengths. (b) Dependence of nonlinear refractive index and third-order susceptibility of monolayer IF-WS<sub>2</sub> NPs on wavelength.

nanosheet [23], so the influence of the solvent on the final third-order nonlinearity can thus be excluded. As shown in Fig. 3(b), the third-order susceptibility  $\chi^{(3)}_{monolayer}$  for monolayer WS<sub>2</sub> varies slightly around 720–800 nm. Both the nonlinear refractive index and third-order susceptibility  $\chi^{(3)}_{monolayer}$  obtained by the SSPM experiment are listed for an explicit comparison (Table 1). Regarding the planar 2D WS<sub>2</sub>, the introduced additional freedom by curved features plays an encouraging role in boosting up the nonlinear characteristics. On the other hand, compared with other 2D materials, such as black phosphorus (BP), IF-like WS<sub>2</sub> NPs with superior  $n_2$  and  $\chi^{(3)}_{monolayer}$  highlight a better idea of improving nonlinear optical properties.

Figure 2(b) briefly demonstrates the evolution of the diffraction pattern. The concentric rings pour out from the center. The diffraction pattern approaches the maximum geometric size within ~0.5 s (Fig. 4). Subsequently, both the horizontal and vertical diameters of the rings collapse and reach a steady state after ~2.8 s and ~4.5 s, respectively (Fig. 4). In contrast, the vertical diameter shrinks to half of the maximum one, while the horizontal diameter only compresses to 82% of the maximum one. The third-order nonlinearity is estimated when the number of rings becomes stable.

The distortion of diffraction rings is mainly attributed to the change of local material concentration induced by the non-axis-symmetrical thermal convection [43,44]. When the laser is incident upon the dispersions, the temperature surrounding the

laser beam becomes asymmetrical, as the temperature gradient above the laser beam rises while it remains nearly stationary below the laser beam. As the non-axis-symmetrical thermal conduction increases [45], WS $_2$  NPs in the upper part of the dispersions are precipitated into the lower part, resulting in a smaller density of WS $_2$  NPs in the upper half of the dispersions, and then a reduced  $N_{\rm eff}$ , naturally, with a reduced  $n_2$ . Therefore, the lower-half dispersions have a relatively stronger nonlinear optical response, leading to the vertical collapse of the SSPM diffraction rings. Notably, the vertical deformation of SSPM rings is of great significance for the study of the photorefractive index change of IF-like WS $_2$  NPs.

The maximum value of the vertical radius of the outermost ring and its half-cone angle are denoted by  $R_H$  and  $\theta_H$ , respectively. The half-cone angle can be written as  $\theta_H = \lambda/2\pi(\mathrm{d}\Delta\varphi/\mathrm{d}r)_{\mathrm{max}}$ , which can be further simplified, for a Gaussian beam, to be  $\theta_H \approx n_2 IC$ , where  $C = [-(8IrL_{\mathrm{eff}}/\omega_0^2) \times \exp(2r^2/\omega_0^2)]_{\mathrm{max}}$  with  $r \in [0, +\infty)$  being a constant. The distortion angle can be expressed as  $\theta_D \approx \Delta n_2 IC$ , where  $\Delta n_2$  is the nonlinear refractive index change caused by intensity variation. Eventually, the change ratio of the nonlinear refractive index can be calculated [39,43].

An increased incident intensity induces a more obvious distortion. Figure 5 exhibits the relationship between incident intensity and  $\Delta n_2/n_2$  at different wavelengths. To a certain extent, the linear regulation of the refractive index change of the material can be achieved by adjusting the intensity of the applied optical field. Nevertheless, the distortion ratio

Table 1.  $n_2$  and  $\chi^{(3)}_{monolaver}$  for Different 2D Materials Obtained by SSPM

2D Materials	$n_2  (\mathrm{cm}^2/\mathrm{W})$	$\chi_{\text{monolayer}}^{(3)}$ (esu)	Laser Wavelength (nm)	References
Graphene	$2.5 \times 10^{-5}$	10 <sup>-7</sup>	532, CW	[20]
$MoS_2/WS_2/MoSe_2$	~10 <sup>-7</sup>	$10^{-9}$	488, CW	[23]
BP 2, 2, 2	~10 <sup>-5</sup>	~10 <sup>-8</sup>	350-1160, pulse	[38]
SnS	~10 <sup>-5</sup>	$\sim 10^{-10}$	532/633, CW	[39]
Antimonene	$\sim 10^{-5}$	$\sim 10^{-8}$	405/785/1064, CW	[40]
$Ti_3C_2T_x$	~10 <sup>−4</sup>	~10 <sup>-7</sup>	457/532/671, CW	[41]
Te	~10 <sup>-5</sup>	/	457/532/671, CW	[42]
WS <sub>2</sub> NPs	$\sim 10^{-5}$	~10 <sup>-7</sup>	720-800, pulse	This work

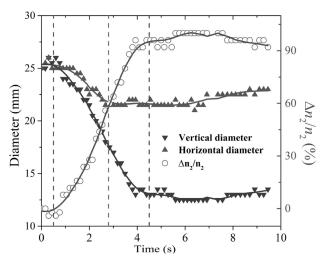
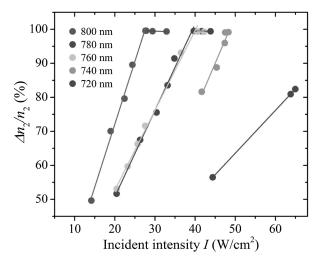


Fig. 4. Evolution of the diameter of the outermost SSPM ring along the vertical and horizontal directions and  $\Delta n_2/n_2$  at  $\lambda=800$  nm.



**Fig. 5.** Dependence of  $\Delta n_2/n_2$  on the incident intensity at different wavelengths.

cannot be infinitely large due to the limitation  $\theta_D < \theta_H$ . When the incident intensity reaches the wavelength-dependent threshold of approximately 30-40 W/cm<sup>2</sup>, the distortion ratio is prone to saturation (Fig. 5). Even so, the Kerr effect itself is not saturated. Since the saturation of the distortion phenomenon is mainly influenced by the non-axis-symmetrical thermal convection, a vertically rising temperature gradient causes WS<sub>2</sub> NPs to continuously sink below the laser beam. After a period of thermal convection, when the density of WS<sub>2</sub> NPs above the laser beam is infinitely close to zero, the upper part of the diffraction rings gradually approaches complete collapse.

As shown in Table 1, only the third-order nonlinear performance of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene exceeds the counterparts of the proposed WS<sub>2</sub> NPs. However, the underlying mechanism here is different from those observed in  $Ti_3C_2T_x$  MXene with a narrow direct bandgap [41]. Because of the multiple layers in WS<sub>2</sub> NPs, no photoluminescence (PL) emission is observed in our experiment [46,47]. Therefore, no interband transition occurs.

The electrons are delocalized by the polarized incident field. The nonlinear refractive index can be estimated by  $\chi^{(3)} \approx Ne^4/\varepsilon_0 m^3 \omega_{e0}^6 d^2$ , where e is the element charge,  $\varepsilon_0$  is the vacuum permittivity, N is the density of electrons of the material,  $\omega_{e0}$  is the oscillation frequency of electrons,  $\omega_{e0} =$  $me^4/32\pi^2\varepsilon_0^2\hbar^3$ , d is the lattice constant, and m is the effective mass of the conduction electron [1]. If d is identified with the Bohr radius  $a_0 = 4\pi\varepsilon_0 \hbar^2/me^4$ , we obtain that  $\chi^{(3)} \propto m^{-7}$ . Due to the distortion and curved features in WS2 NPs, the effective mass of electrons in IF-like WS2 NPs is speculated to reduce in comparison with the counterparts in planar 2D materials [25]. Therefore, the reduced effective mass of electrons will contribute to a portion of the enhancement in  $n_2$ and  $\gamma^{(3)}$ .

The mechanism of the SSPM phenomenon in WS<sub>2</sub> NPs dispersion is essentially an appearance of intensity-dependent change in the refractive index. In principle, the thermal effect can only play a crucial role when the pulse duration is longer than tens of picoseconds. Therefore, the thermal contribution plays a non-dominated role in  $n_2$  and  $\chi^{(3)}$  enhancement under the illumination of the fs pulse source. Nevertheless, its contribution may be comparable to the contribution of the reduced effective mass of electrons. The electrons and holes generated by photoexcitation will drift in directions that are antiparallel and parallel to the electric field, respectively, resulting in polarized WS2 NPs. Initially, an arbitrary angle related to the interaction energy exists between the direction of the WS2 NPs polarization and the laser-induced electric field. As interaction energy is minimized, WS2 NPs are reoriented and aligned. The isotropy of the carriers in each particle appears as a kind of coherence that contributes to the macroscopic SSPM phenomenon. While it has another explanation, the gap-dependent SSPM can be regarded as a purely coherent third-order nonlinear optical process, which is generated from the nonlocal ac electron coherence within the sample [35]. Since each WS<sub>2</sub> NP is mimicked as a separated domain containing multiform carriers, anisotropic domains are reoriented to alignment attributed to the torque produced by interior electron coherence influenced by an external electromagnetic field and finally polarized. The dielectric polarization caused by the electron coherence effect can be regarded as the collective behavior of a large number of electrons within the sample. Similarly, the polarization induced by the drift of photoexcited carriers (holes) can also be considered as a collective behavior of carriers.

Recently, it was demonstrated that second-harmonic generation can be actively controlled via the generation of photocarriers in monolayer MoS2 using ultrashort pulses, which enables a promising time-resolved approach to characterize the second-order nonlinear response [48]. A similar approach is also promising for extension into unveiling the detailed physical mechanism of the enhanced third-order nonlinear properties of  $WS_2$  NPs.

#### 4. CONCLUSION

In conclusion, a novel type of IF-like WS<sub>2</sub> NPs is successfully synthesized using the hard template method with a diameter of 26.5 nm. By characterizing the nonlinear refractive index  $n_2$ and third-order susceptibility  $\chi^{(3)}$  using the SSPM method with a visible fs pulse laser, we obtain  $n_2 \sim 10^{-5}~{\rm cm^2/W}$  and  $\chi^{(3)} \sim 10^{-7}$  esu, which are orders stronger than the counterparts of planar 2D materials. In addition, the enhanced third-order nonlinear response can be controlled flexibly by varying the excitation wavelength and incident intensity, which is beneficial for all-optical devices. Therefore, IF-like 2D materials will enrich the optical materials with superior efficiencies, and are endowed with promising potentials in photonic integration applications.

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